# **Interlayer Structure and Molecular Environment of Alkylammonium Layered Silicates**

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Transmission Fourier transform infrared spectroscopy (FTIR) was used to probe the interlayer structure and phase state of intercalated alkylammonium silicates by monitoring frequency shifts of the **CH2** stretching and scissoring vibrations as a function of interlayer packing density, chain length, and temperature. **A** wide range of molecular environments varying from solidlike to liquidlike was found. In general, as the interlayer packing density or chain length decreases or the temperature increases, the chains adopt a more disordered, liquidlike structure. In intermediate cases, the chains are neither solidlike nor liquidlike but exhibit a liquid-crystalline character.

## **Introduction**

Alkylammonium derivatives of mica-type layered silicates (MTS) are employed in a wide variety of industrial and scientific applications. Examples include rheological control agents in paints and greases, $^{1,2}$  adsorbents for the treatment of contaminated waste streams. $3,4$  and models for studying chain aggregation in biomembranes. $5$  More recently, alkylammonium silicates have been used as artificial membranes,  $6$  chemical sensors,  $7.8$  host structures for direct melt intercalation of polymers,<sup>9</sup> and reinforcements in polymer matrix composites.<sup>10,11</sup> Last, they have been found to mediate the formation of mesoporous silicates.<sup>12</sup> In all of these applications, the behavior and properties of the hybrids depend largely on the structure and the molecular environment of the organic interlayer.

Traditionally, structural characterization has been limited to determining the orientation and arrangement of the alkyl chains based on X-ray diffraction analysis (XRD). Depending on the packing density, temperature, and chain length, the chains are thought to lie either parallel to the host layers forming lateralmono- or bilayers or radiate away from the surface forming extended (paraffin-type) mono- or bimolecular arrangements $13-16$ 

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Figure 1. Alkyl chain aggregation in mica-type silicates: (a) **lateral monolayer; (b) lateral bilayer; (c) paraffin-type monolayer; (d) paraffin-type bilayer. Adopted from ref 15.** 

(Figure 1). In some cases, individual chains are thought to adopt a hybrid arrangement with both lateral and paraffin-type segments leading to a pseudotrilayer. Such idealized structures, based almost exclusively on *all-trans*  segments are potentially misleading, since they fail to convey the most significant structural characteristic of aliphatic chains-the capacity to assume an enormous array of configurations because of the relatively small energy difference between *trans* and *gauche* conformers *(0.6* kcal/mol, **2.5** kJ/mol) **.I7** 

For illustrative purposes let us consider the case where the chains are thought to be in an extended (paraffintype), tilted arrangement. An alternative arrangement based on a disordered chain configuration containing *numerousgauche* conformers would **also** be consistent with the observed gallery height (Figure **2).** The different arrangements, however, although indistinguishable by X-ray, lead to much different interlayer structure and molecular environment. Determining the interlayer struc-

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Figure **2.** Different chain arrangements leading to the same gallery height: (a) tilted, *all-frons* chains and (b) chains with numerous gauche conformers. Open circles *(O)* represent CH<sub>2</sub> segments while cationic head groups are represented by filled circles **()**.

ture solely on X-ray diffraction measurements is thus inadequate.

Investigators in many fields including colloidal and  $interface science, <sup>18,19</sup> polymer science, <sup>20</sup> and biology<sup>21</sup> have$ extensively used Fourier transform infrared spectroscopy (FTIR) to probe the structure and organization of molecules near interfaces and in colloids, aggregates, and bimolecular films. They have found that the frequency, width, height, and integrated intensity of the  $CH<sub>2</sub>$  and  $CH<sub>3</sub>$  infrared adsorption bands are sensitive to the gauche/ trans conformer ratio of the chains **as** well **as** the intermolecular interactions between the chains.<sup>18</sup> Thus they can he used to monitor structural rearrangements resulting from changes in packing density, temperature, pressure, and mechanical stress.18.2z

In this study, transmission FTIR is used in conjunction with XRD to probe the interlayer structure of intercalated alkylammonium silicates. Using various organically modified silicates, we have monitored the frequency shift of the asymmetric  $CH<sub>2</sub>$  stretching and scissoring vibrations as a function of interlayer packing density, chain length, and temperature. The FTIR measurements provide for the first time direct information on the structure of the organic interlayer as well **as** a means to relate the intercalated state to a corresponding bulk phase. The additional information garnished from these studies provide a more realistic description of the interlayer environment of the hybrids and could potentially improve our understanding of their properties.

#### Experimental Methods

Materials. Na+-montmorillonite with an exchange capacity of *80* mequiv/100 g (WM) was obtained from the University of Missouri Source Clay Repository (SWy-1). A higher exchange capacity Na<sup>+</sup>-montmorillonite (100 mequiv/100 g, SM) and Li<sup>+</sup>fluorohectorite (150mequiv/100g,FH) weredonated bySouthem Clay Products, Inc., and Corning, Inc., respectively. Dioctadec-<br>yldimethyl ammonium bromide  $(2C_{18}N^+2C_1Br$ , Kodak) and a series of primary alkyl amines  $(C_nH_{2n+1}NH_2, n = 6, 9, 10, 11, 12,$ 13, 14, 16, and 18, Aldrich Chemical) were used **as** received.

**Synthesis.** The organosilicate hybrids  $(FH-2C_{18}, SM-2C_{18},$  $WM-2C_{18}$  and  $FH-C_n$ ) were synthesized by a cation-exchange reaction between the layered silicates and excess alkylammonium salt (twice the exchange capacity of the host). The quaternary ammonium cation or the primary amine were dissolved in a *<sup>50</sup>* 50 mixture of ethanol and deionized H<sub>2</sub>O at 50-70 °C. In the case of primary amines an equivalent amount of HCI was also addedtothesolution. Alwt *7%* **aqueoussuspensionofthelayered**  silicate wasadded **totheaikylammoniumsolutionand** themixture was stirred for 5-6 h at 50-70 °C. The cation-exchanged silicates were collected by filtration and subsequently washed with a mixture of hot ethanol and deionized  $H_2O$  until an AgNO<sub>3</sub> test indicated the absence **of** halide anions. The filter cake was dried at room temperature, ground, and further dried at 70-80 °C under vacuum for at least 24 h. The final powders were stored under vacuum with  $P_2O_5$  desiccant.

Characterization. The interlayer packing densities were estimated thermogravimetrically (TGA) from the percent weight loss associated with decomposition of the interlayer organics.<br>The analyses were carried out in air on a DuPont 951 Thermo-<br>gravimetric analyzer at  $10^{\circ}$ C/min. These results were selectively verified with elemental analysis (Galbraith Laboratories). The absence of either **CI** or Br indicates that the hybrids are stoichiometric and contain no excess of alkylammonium ions.<br>FTIR spectra were collected using a Galaxy 2020 spectrometer

with a nominal resolution of 4 cm<sup>-1</sup>. Spectra were obtained from KBr pellets or films confined between two NaCl crystals mounted in a thermostatically controlled cell (Spectra Tech, Inc.). Care was taken to press all KBr pellets at the same conditions to minimize any effect of pressure on peak frequencies. Solution spectra in chloroform were obtained using a Perkin-Elmer demountable liquid cell. For each spectrum 64 interferograms were collected and coadded. Peak frequencies were determined using zero-point determination of the first derivative or center of gravity method and were reproducible with successive experiments to at least  $\pm 0.1$  cm<sup>-1</sup>.<sup>23</sup>

**XRDspectrawerecollectedonaScintagInc.8-Bdiffractometer**  equipped with an intrinsic germanium detector system using either CuK $\alpha$  or CrK $\alpha$  radiation. Differential scanning calorimetry (DSC) was performed using a TA Instruments cell at  $10 °C/min$ under  $N_2$ .

# Results and Discussion

Effect of Packing Density. The effect of packing density on the interlayer structure was examined by intercalating dioctadecyldimethyl ammonium  $(2C_{18})$  into WM, SM, and FH. As expected the interlayer packing density parallels the exchange capacity of the host increasing from WM to FH (Table 1). X-ray diffraction patterns of the hybrids show several harmonics with a

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**Figure 3. Selected regions** of **FTIR absorbance spectra** of **WM-** $2C_{18}$ , SM-2C<sub>18</sub>, and FH-2C<sub>18</sub> hybrids displaying various methyl **and methylene modes. The spectra were obtained at room temperature.** 

**Table 1. Summary of XRD and FTIR Data for MTS-2Cls Hybrids** 

	$WM-2C_{18}$	$SM-2C_{18}$	$FH-2C_{18}$
aliphatic ions/ $\operatorname{Si}_8O_{20}$ unit	0.65	0.80	$1.2\,$
gallery height (nm) <sup>a</sup>	1.43	1.57	2.85
layer area/molecule (nm <sup>2</sup> )	0.72	0.58	0.39
layer area/chain $(nm^2)$	0.36	0.29	0.20
$\nu_{\rm as}$ , CH <sub>2</sub> (cm <sup>-1)b</sup>	2924.7	2923.6	2919.6
$\delta$ , CH <sub>2</sub> (cm <sup>-1)b</sup>	1468.1	1468.3	1468.9

 $a_{001} - 0.95$  nm (host layer thickness).  $b$  The FTIR spectra were **obtained at room temperature.** 

primary repeat unit,  $d_{001}$ , of 2.38, 2.52, and 3.80 nm for  $WM-2C_{18}$ , SM-2C<sub>18</sub>, and FH-2C<sub>18</sub>, respectively.

To directly probe the molecular conformation of the intercalated chains and provide a better insight to the interlayer structure, FTIR spectroscopy was used. The strongest aliphatic absorption bands observed in the MTS-2C18 hybrids are the methylene modes between 3000 and  $2800 \text{ cm}^{-1}$  and between 1520 and 1400  $\text{cm}^{-1}$  (Figure 3). The bands at  $\sim$  2920 and 2850 cm<sup>-1</sup> arise from the CH<sub>2</sub> asymmetric,  $v_{\text{as}}(\text{CH}_2)$ , and symmetric,  $v_{\text{a}}(\text{CH}_2)$ , stretch, respectively. The band around  $1470 \text{ cm}^{-1}$  is due to the  $CH_2$  bending (scissoring) vibration,  $\delta CH_2$ ).<sup>24</sup> The peak frequency of these modes along with packing density, gallery height, and area available per guest molecule for  $WM-2C_{18}$ , SM-2C<sub>18</sub>, and FH-2C<sub>18</sub> are summarized in Table 1.

As the interlayer packing density increases,  $\nu_{as}(CH_2)$ shifts from 2924.7 to 2923.6 to 2919.6 cm<sup>-1</sup> for WM-2C<sub>18</sub>,  $SM-2C_{18}$ , and  $FH-2C_{18}$ , respectively. Even though the frequency shifts appear to be relatively small, they are rather significant considering that they are reproducible to a least  $\pm 0.1$  cm<sup>-1</sup>. In general, the frequency and width of  $\nu_{\rm ss}(\rm CH_2)$  are sensitive to the *gauche/trans* conformer ratio and the packing density of methylene chains.<sup>25-29</sup> The band shifts from lower frequencies, characteristic of highly ordered *all-trans* conformations, to higher frequencies and increased widths **as** the number of *gauche*  conformations along the hydrocarbon chain (chain disorder) increases.28 For example, the methylene chains in the *all-trans* ordered state of crystalline  $2C_{18}N+2C_1Br$ exhibit a band at 2917.8 cm-l. The band shifts to 2928.9 cm-1 when the chains are in a liquidlike environment as in  $2C_{18}N+2C_{1}Br/CHCl<sub>3</sub>$  solution.

For  $FH-2C_{18}$ , an extended monolayer with chains normal to the layers **or** a tilted bilayer (Figure 1) are consistent with the measured gallery height. The proximity of  $v_{\text{as}}(CH_2)$  to that observed for crystalline  $2C_{18}N^+2C_1Br^$ indicates that a monolayer of predominantly extended, *all-trans* chains is present. In such a configuration there is a perfect match between the cross-sectional area of the extended molecules  $(0.40 \text{ nm}^2)^{30}$  and the available area/ guest molecule (0.39 nm2). The presence of extended chains normal to the surface is further supported by the observed gallery height (2.85 nm), which is also consistent with a monolayer of fully stretched chains  $(\sim 2.6 \text{ nm}^{31,32})$ . A bimolecular arrangement (Figure Id) with fully stretched chains tilted  $33^{\circ}$  with respect to the surface seems unlikely. since the large area/molecule available (0.80 nm2) would favor a disordered configuration due to the tendency of the chains to maximize their conformational entropy, which would lead to much higher  $\nu_{\text{as}}(CH_2)$  than observed.

As the chain packing density progressively decreases and the available surface area per guest molecule increases, the gallery height decreases to 1.57 and 1.43 nm in SM- $2C_{18}$  and WM- $2C_{18}$ , respectively. The decrease in gallery height is accompanied by similar shifts in the  $\nu_{\text{as}}(CH_2)$  to higher frequencies consistent with chains with increased *gauche* conformers. With decreasing packing density, the chains are no longer fully stretched in **an** *all-trans*  conformation but they progressively adopt a more disordered structure (Figure 2b). Again, a tilting arrangement containing *all-trans* chains, although consistent with the XRD measurements, seems unlikely in view of the IR data.

Aggregation of chains into kink and gauche blocks containing *gauche* conformers has been previously proposed by Lagaly to account for the temperature dependence of gallery height in high-temperature phases of bimolecular paraffin-type hybrids.<sup>5</sup> We assert that the presence of gauche conformations is more ubiquitous while *all-trans* segments are present only under special circumstances (i.e., when the cross-sectional area of the intercalated molecules is about the host area/guest molecule). Furthermore, to the best of our knowledge this is the first direct experimental confirmation for the presence of *gauche* segments in intercalated systems.

Using the frequency shifts **as** a guide, we can also determine the phase state of the interlayer. Increased

<sup>(24)</sup> In addition to the asymmetric and symmetric CH<sub>2</sub> stretching and **CH2 scissoring vibrations monitored in the present study, numerous other methylene and head group modes can be utilized in structural investiga**tions of the tails of common end-functionalized aliphatic compounds.<sup>18,28</sup> **Many of these modes though were weak or masked by the silicate host.** 

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chain order leads to more efficient packing and increased interchain contacts resulting in more cohesive van der Waals interactions between the chains and a greater interlayer solidlike character. Conversely, increased *gauche*  conformations (increased disorder) leads to a more liquidlike state.<sup>19</sup> On the basis of the  $CH<sub>2</sub>$  stretching band the chains in  $FH-2C_{18}$  appear to be in a solidlike environment and become progressively more liquidlike in SM- $2C_{18}$  and WM- $2C_{18}$ .

Additional structural and phase-state information may be gathered from the location and shape of the CH2 bending mode,  $\delta$ (CH<sub>2</sub>). This band is sensitive to interchain interactions and the packing arrangement of the chains with frequencies ranging from 1466 to 1472 cm<sup>-1</sup>.<sup>20,27-29</sup> In general higher frequencies  $({\sim}1472 \text{ cm}^{-1})$  indicate "ordering" of the methylene chains in an *all-trans* crystalline state. Conversely, lower frequency  $(\sim 1466 \text{ cm}^{-1})$ , band broadening, and decreasing intensity indicate a decrease in interchain interactions and an increase in chain motion which is normally associated with a liquid state.<sup>28</sup>

All three  $MTS-2C_{18}$  hybrids display an absorption between 1468 and 1469 cm-1 with the lower frequency corresponding to the lower exchange capacity host (Figure 3, Table 1). Even at the lowest packing density (WM), we do not observe the broadening and decreased intensity associated with fully disordered, liquidlike chains.<sup>28,29</sup> The observed absorption around 1468 cm-l is characteristic of a partially ordered phase where the chains are mobile while maintaining some orientational order.<sup>28</sup> A similar effect has been observed in the liquid-crystalline state of lipids, whose spectra **also** exhibit a band at 1468 cm-1.28

On the basis of the above, we suggest that even in the low packing density hosts (WM and SM) the interlayer is not completely disordered (liquidlike), but the chains retain some orientational order similar to that in a liquidcrystalline (LC) state. This assertion is further supported by the proximity of the  $\nu_{\text{as}}(CH_2)$  for the two montmorillonites to that of liquid crystalline  $2C_{18}N+2C_{1}Br$  (refs 31 and 32) (2923.8 cm<sup>-1</sup> at  $T = 80$  °C) and not to the frequency characteristic of completely disordered chains of the same salt in solution (2928.9 cm<sup>-1</sup>). Therefore, in contrast to bulk states of matter where the positional and orientational order of the molecules is solely dictated by the molecules themselves, the intercalated molecules retain some orientational order imposed by the physical presence of the silicate layers and the packing density requirements that maintain charge neutrality.

**Effect of Chain Length.** The dependence of interlayer structure on chain length has also been investigated. Figure 4 compares  $\nu_{\text{as}}(CH_2)$  and gallery height for a series of organosilicate hybrids  $(FH-C_n)$  of varying alkyl chain length, C,, but similar packing density. **As** the chain length increases,  $v_{\text{as}}(CH_2)$  shifts stepwise from 2932 cm<sup>-1</sup> for FH- $C_6$  to 2921.1 cm<sup>-1</sup> for FH-C<sub>18</sub>. The frequency shifts in  $v_{\text{as}}(CH_2)$  are mirrored by similar step decreases in the gallery height (Figure 4) suggesting that they are interrelated.

On the basis of conventional models $^{5,13-16}$  which relate the interlayer structure to chain length and packing density, we would predict a lateral monolayer for  $FH-C_6$ , lateral bilayer for  $FH-C_9$  through  $FH-C_{12}$  and eventually pseudotrilayer to paraffin-type for  $FH-C_{13}$  to  $FH-C_{18}$  (see Figure 1). The attractive interactions between the silicate surface and the intercalated molecules and the geometric constraining effect on these molecules most likely force



Figure 4.  $\nu_{\text{as}}(CH_2)$  and gallery height as a function of chain length for  $FH-C_n$  measured at room temperature.



**Figure 5.** Room-temperature FTIR spectra for FH-C<sub>n</sub> showing changes in the  $\delta$ (CH<sub>2</sub>) mode with decreasing chain length.

then to order in quasi-discrete layers.30 The large variation in  $\nu_{\text{as}}(CH_2)$  for the FH-C<sub>n</sub>, however, suggests that the chains are in a wide range of molecular environments which is not obvious or implicit in the previously proposed idealized structures. Recall that shifts in the  $CH<sub>2</sub>$  stretching bands largely reflect changes in chain conformation, with the frequency shifting from lower frequencies for ordered, all*trans* chains, to higher frequencies **as** the chain disorder *(gaucheltrans* conformer ratio) increases. Conversely, interchain interactions dominate the  $CH<sub>2</sub>$  bending vibrations. Thus, if the chains in  $FH-C_n$  were to adopt the structures described above (see Figure l), we would expect little variation in the CH2 stretching frequencies in contrast to what we have observed.

By comparing the frequency of FH-C18 to bulk  $C_{18}NH<sub>3</sub>$ <sup>+</sup>Cl<sup>-</sup>, we find that the chains in the hybrid are in an LC state. This finding is consistent with our previous assignments, since the layer area/chain for both  $FH-C_{18}$  $(0.39 \text{ nm}^2)$  and WM-2C<sub>18</sub>  $(0.36 \text{ nm}^2)$  is similar and in the latter the chains are **also** in an LC state. Incidentally, the gallery height for the two hybrids is also comparable (1.33 and 1.43 nm for  $FH-C_{18}$  and WM-2 $C_{18}$ , respectively). As



Figure 6. Alkyl chain aggregation models for FH-C<sub>n</sub>. As the number of carbon atoms increases, the chains adopt a more ordered structure. For the shortest lengths (a), the molecules are effectively isolated from each other discrete layers form with various degrees of inplane disorder and interdigitation between the layers. For longer lengths (c), interlayer order increases leading to a LC environment. Open circles *(0)* represent CH2 segments while cationic head groups are represented by filled circles *(0).* The top silicate layer has been left transparent to improve the perspective of the interlayer.

the chain length decreases and the available area per molecule increases the chains adopt a progressively more disordered structure and eventually become liquidlike for  $n < 12$ . In FH-C<sub>6</sub>, due to the short chain length, the chains are rather isolated such that the stretching frequency resembles that of chains in agaseous state. The transition to a more liquidlike environment with decreasing chain length is further supported by decreasing intensity, peak broadening, and a shift to lower frequencies of the  $\delta$ (CH<sub>2</sub>) mode (Figure 5). A schematic representation of the proposed structures for the  $FH-C_n$  series is shown in Figure 6.

The proposed structures are consistent with recent molecular dynamic (MD) simulations of tethered and free alkane molecules either on free surfaces or confined between parallel plates. In general the physical presence of the impenetrable interface induces both normal and in-plane ordering of the chains. $33-35$  The fraction of gauche conformers, however, although generally less than that observed for the liquid alkanes **(30%),** is still quite substantial. In addition, some interdigitation of chain segments between those closest to the interface and subsequent layers has been observed. Of particular interest are simulations of Cg alkane molecules tethered to a free silicate surface with an area comparable to the FH-C<sub>n</sub> series  $(0.41 \text{ nm}^2 \text{ per chain})$ . <sup>36</sup> The simulations show that at room temperature the chain gauche conformer content is **30%,** which is comparable to that of liquid alkanes and in excellent agreement with our assignment for  $n < 12$ .

Effect of Temperature. Last, the effect of temperature on the interlayer structure and phase state was examined. Figure 7 shows the temperature dependence of  $\nu_{\text{as}}(CH_2)$  for FH-C<sub>18</sub>. The respective behavior of WM-2C18 is shown in Figure 8. In general, **as** the temperature increases the hand shifts from that characteristic of the LC state to higher frequencies before it finally reaches a plateau with a frequency approaching that of a liquidlike state. The more disordered environment at elevated temperatures is further supported by peak broadening, decreasing intensity, and a shift to lower frequencies of the  $\delta$ (CH<sub>2</sub>) mode, as shown for FH-C<sub>18</sub> in Figure 9. With



**Figure 7.**  $\nu_{as}(CH_2)$  of  $FH-C_{18}$  as a function of temperature: (O) heating; ( $\Delta$ ) cooling. The corresponding DSC trace (10 °C/min in **N2** ambient) is **also** included.



**Figure 8.**  $\nu_{\rm ss}$ (CH<sub>2</sub>) of WM-2C<sub>18</sub> as a function of temperature with the corresponding DSC trace  $(10 °C/min)$  in  $N_2$  ambient).

increasingly elevated temperatures, successive degrees of disorder are introduced in the interlayer leading to a more liquidlike environment. The chain disordering on heating is consistent with MD simulations where the chains are shown to pass gradually from **a** low-temperature solidlike state to a high-temperature liquidlike state over a certain temperature range.36

The interlayer transitions to a more disordered and eventually **a** liquidlike state are in very good agreement with the phase transitions observed in the DSC (Figures 7 and 8). The characteristic transitions in the  $FH$ -C<sub>n</sub> series diminish in intensity and shift to lower temperatures with

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Figure 9. (a, top) FTIR spectra for  $FH-C_{18}$  at different temperatures showing changes in the  $\delta$ (CH<sub>2</sub>) mode. (b, bottom)  $\delta$ (CH<sub>2</sub>) position as a function of temperature determined by the center of gravity method. The corresponding DSC trace  $(10 \degree C)$ min in  $N_2$  ambient) is also included.

decreasing chain length indicating that the interlayer is already in a more liquidlike environment. In FH-C<sub>18</sub> both the IR and the DSC data suggest that there is a transition to another state before the chains adopt a completely disordered, liquidlike state.

The advantages of FTIR over XRD in probing the interlayer structure is evident by comparing changes in frequency and gallery height as a function of temperature. In contrast to the large frequency variation, which implies a wide range of interlayer environments, the gallery height changes very little over the same temperature range (Figure 10). At low temperatures the gallery height is virtually constant. Between 80 and 130 °C a modest increase of about  $\sim 0.08$  nm is observed after which the gallery height plateaus out. The modest increase in the gallery height corresponds to the transition from solid like to liquid like environment.

The LC to liquidlike transition coincides with a dramatic increase in the capacitance of the hybrid.<sup>7,8</sup> This behavior has been attributed to an increase in the interlayer fluidity, which is consistent with a transition to a more liquidlike environment observed in this study. Further evidence for the increased chain mobility comes from monitoring the  $\nu_{\text{as}}(CH_2)$  during cooling (Figure 7). When the interlayer is liquidlike, the CH<sub>2</sub> frequency is practically identical during the heating or cooling cycles but becomes hysteretic on cooling below the liquidlike range. The chains become more sluggish as they adopt a more solidlike character preventing them from reaching instantaneously their equilibrium configuration, consistent with the MD simulations.<sup>36</sup>



Figure 10. Temperature dependence of gallery height for FH- $C_{18}$ 

Similar to the dielectric behavior, the state of the interlayer and the mobility of the chains are expected to control other properties of the hybrids. For example, while in the LC state the chains are expected to resist shear but flow and relax albeit at a slow rate. Furthermore, the interlayer environment will largely determine the adsorption and swelling characteristics of the hybrids. Experiments are currently underway to investigate the interlayer properties and correlate them with the structure and phase state.

## Conclusions

In summary, FTIR spectroscopy in conjunction with X-ray diffraction studies have provided a new experimental insight into the interlayer structure and phase state of intercalated alkylammonium silicates. By monitoring frequency shifts of the asymmetric  $CH<sub>2</sub>$  stretching and bending vibrations, we have found that the intercalated chains exist in states with varying degrees of order. In general, as the interlayer packing density or the chain length decreases or the temperature increases, the intercalated chains adopt a more disordered, liquidlike structure resulting from an increase in the gauche/trans conformer ratio. When the available surface area/molecule is within a certain range, the chains are not completely disordered, but retain some orientational order similar to that in a liquid crystalline state. Since the chains are freer to move around and reorient in the liquidlike state, they can attain equilibrium configuration much faster compared to those in an LC or solidlike state. These findings differ from the proposed conventional models in which the chains adopt a static, all-trans configuration, with a fixed cross-sectional area.

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